

**Mathematical Theory of Solids:  
from Atomic  
to Macroscopic Scales**

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## Fluids and solids: A comparison

Fluids: Fundamental model:  
Navier-Stokes equation

$$\begin{cases} v_t + (v \cdot \nabla)v + \nabla p = \frac{1}{Re} \Delta v \\ \nabla \cdot v = 0 \end{cases}$$

$v(\cdot, t)$  = velocity field at time  $t$ .

# Solids: fundamental model?

Linear elasticity:

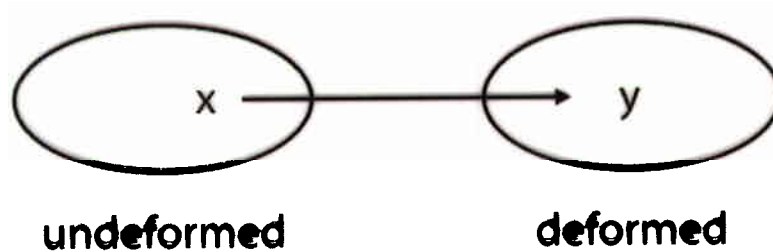
$$\rho \frac{\partial^2 u}{\partial t^2} = (\lambda + \mu) \nabla(\nabla \cdot u) + \mu \Delta u$$

Nonlinear elasticity:

$$\rho \frac{\partial^2 u}{\partial t^2} = \nabla \cdot \nabla_A W(\nabla u)$$

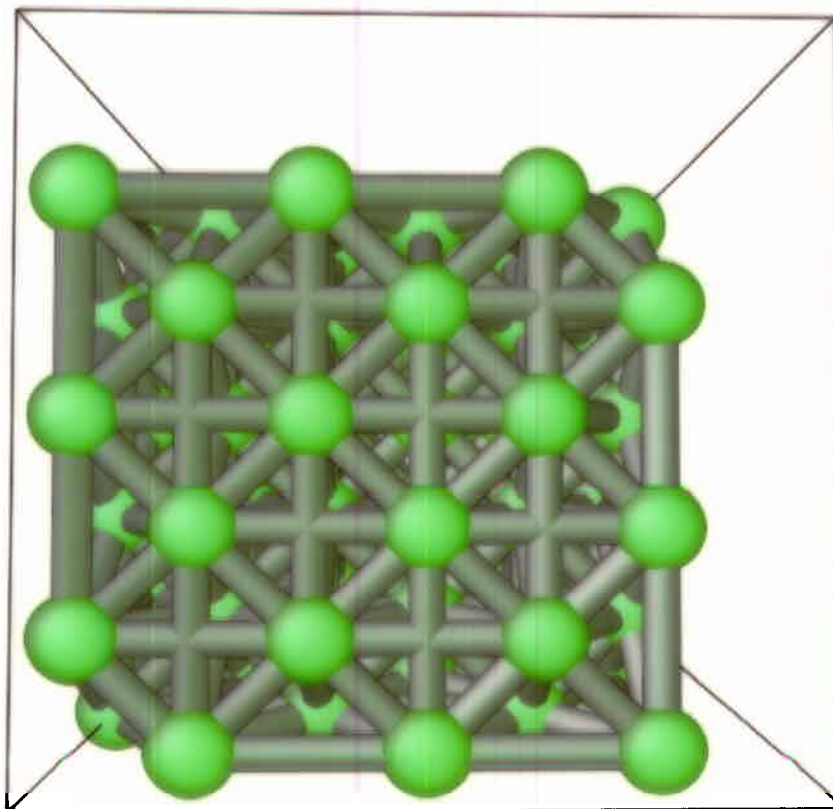
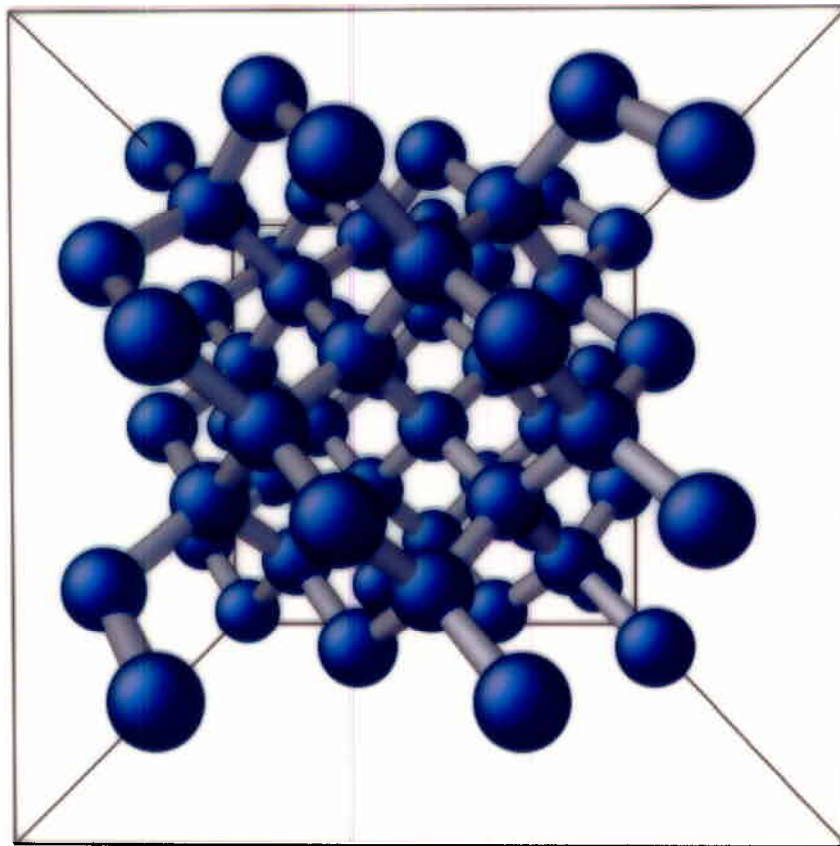
Plasticity

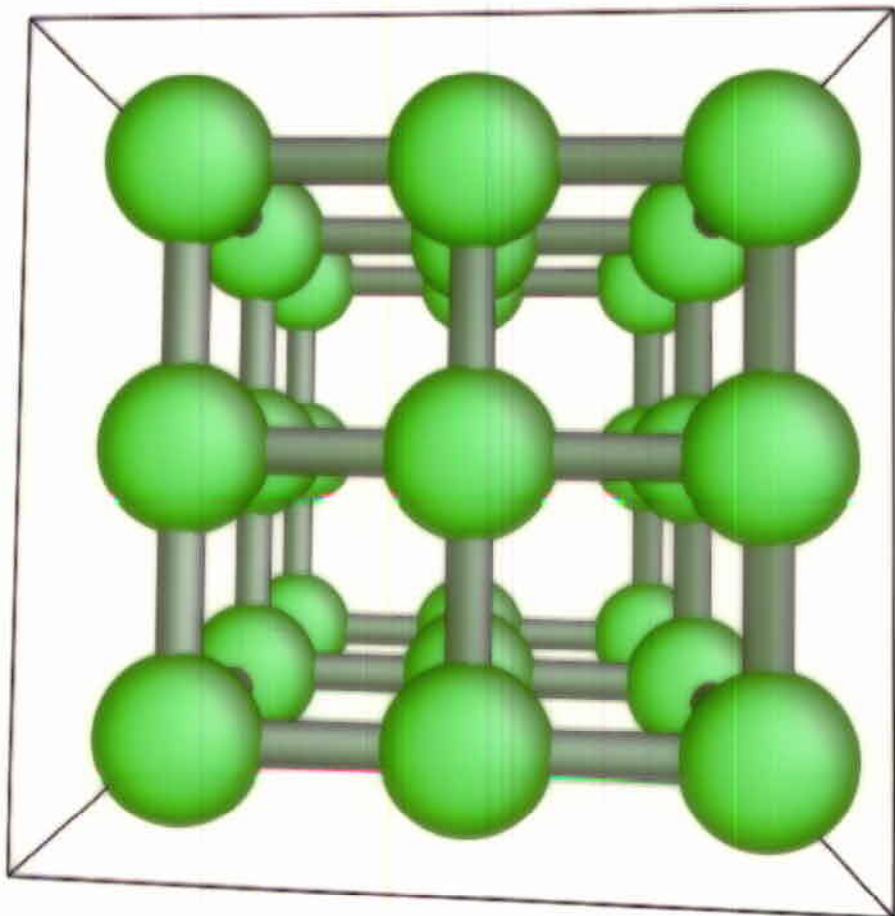
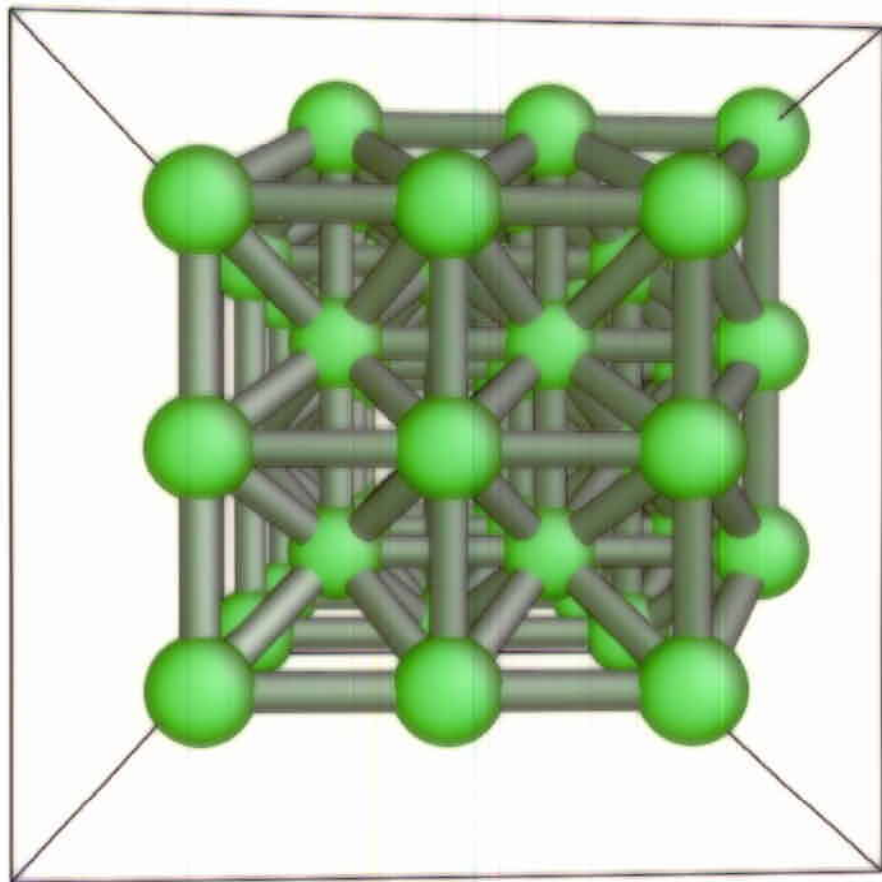
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$$y = x + u(x)$$

$u(x) \equiv$  displacement





# Quantum many-body problem

$$V(R_1, R_2, \dots, R_N) = E^*(R_1, R_2, \dots, R_N) + \sum_{I \neq J} \frac{Z_I Z_J}{|R_I - R_J|}$$

$R_I$  = position of the  $i$ -th nucleus

$Z_I$  = charge of the  $i$ -th nucleus

$E^*$  = ground state energy of the electrons

$$\mathbf{H} = - \sum_i \frac{1}{2} \Delta_{r_i} + \sum_{i < j} \frac{1}{|r_i - r_j|} + \sum_i v(r_i)$$

$$v(r_i) = - \sum_I \frac{Z_I}{|r_i - R_I|}$$

$$\Psi = \Psi(r_1, r_2, \dots, r_M)$$

## **Dirac (1929)**

The underlying physical laws necessary for the mathematical theory of a *large part of physics* and *the whole chemistry* are thus completely known, and the difficulty is only that the exact application of these laws leads to equations much too complicated to be soluble.

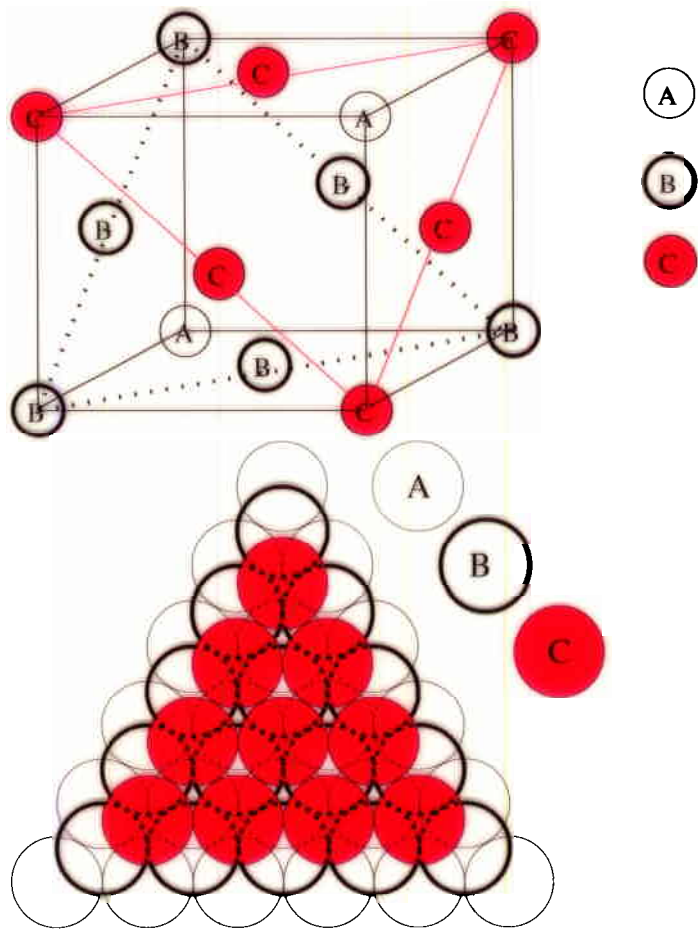
## What do we take as the starting point?

1. Quantum many-body
2. Simplified quantum models (density function theory, tight-binding, etc.)
3. Classical potential
  - Lennard-Jones
  - Embedded Atom Model (EAM)

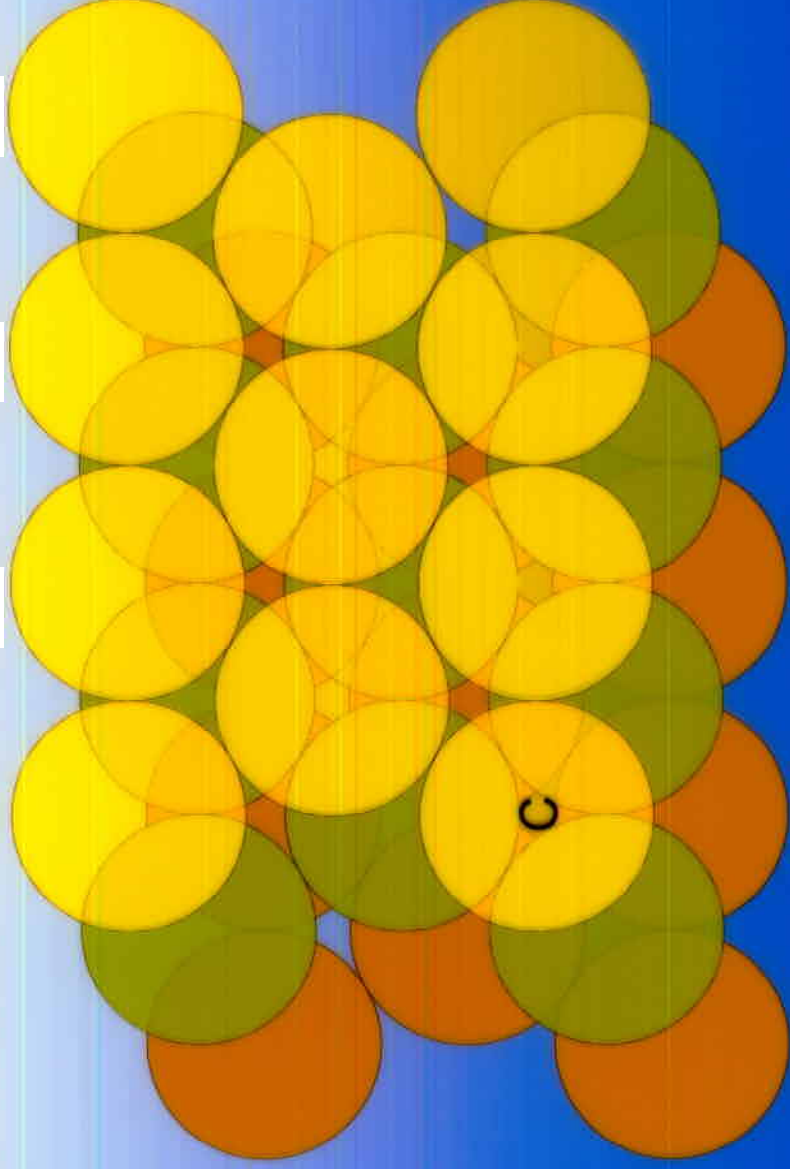
$$V(R_1, R_2, \dots, R_N) = \sum_{I \neq J} V_0(|R_I - R_J|)$$

$$V_0(r) = 4\epsilon \left( \left( \frac{\sigma}{r} \right)^{12} - \left( \frac{\sigma}{r} \right)^6 \right)$$



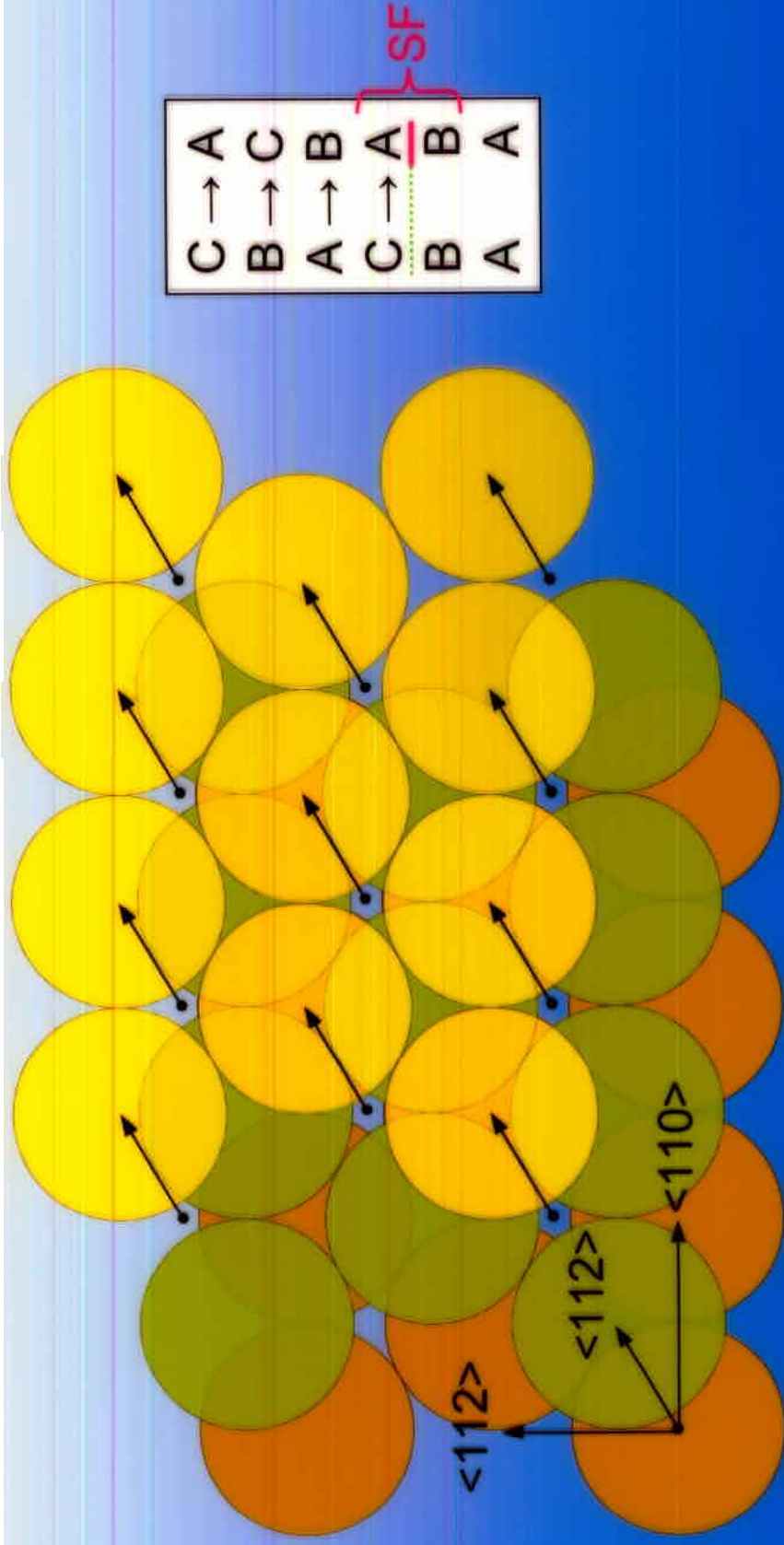


# FCC Stacking

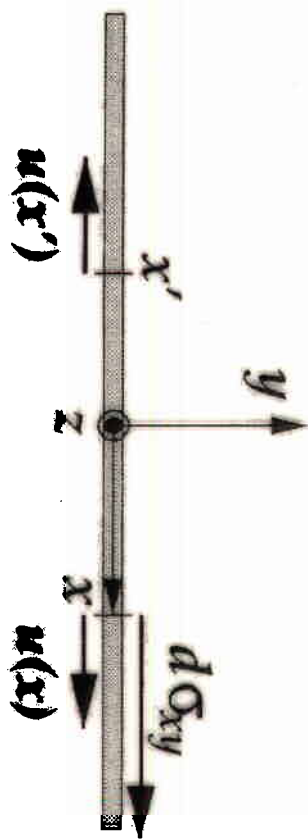
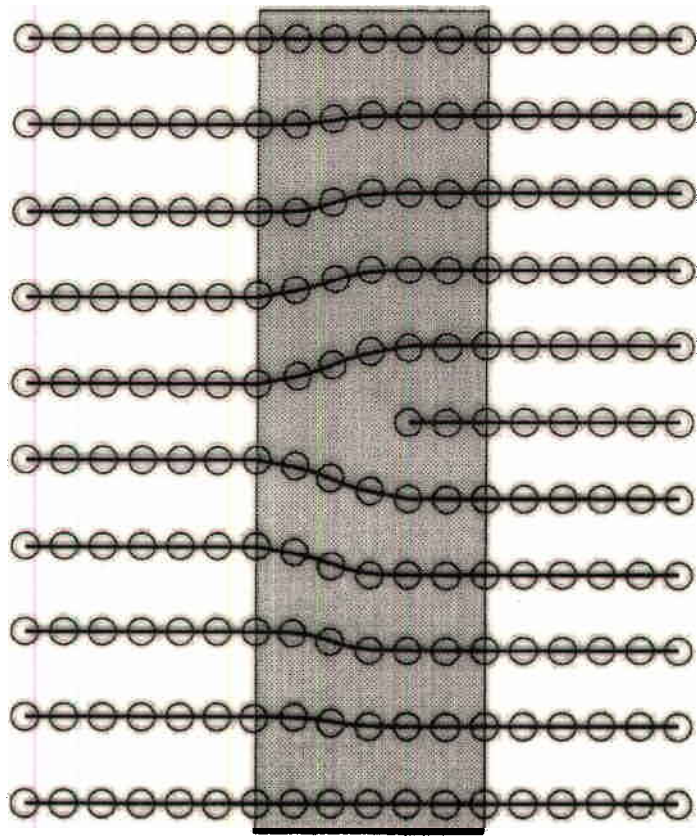


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# Intrinsic Stacking Fault (SF)



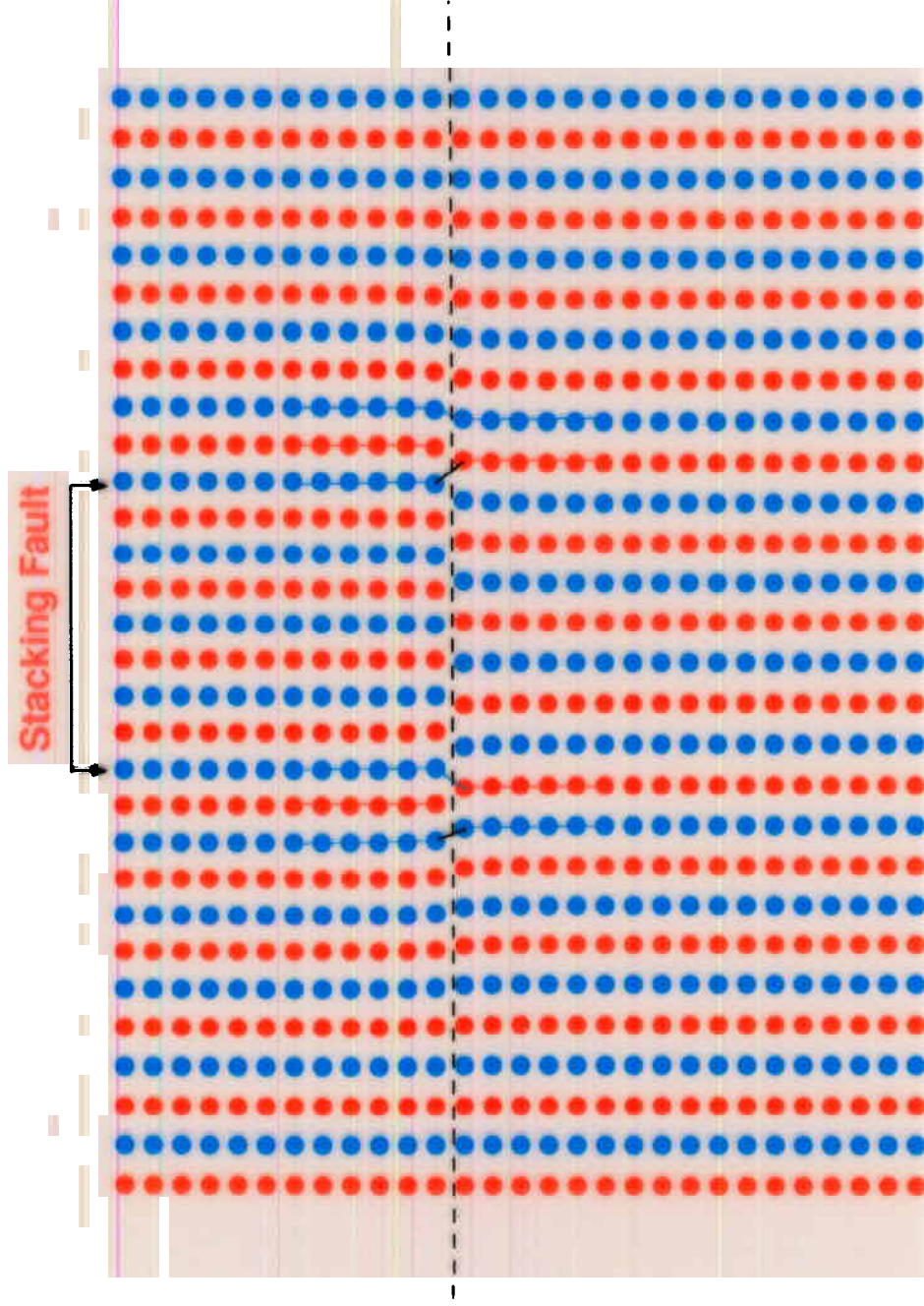
$$\gamma_{sf} = \text{Stacking fault energy}$$



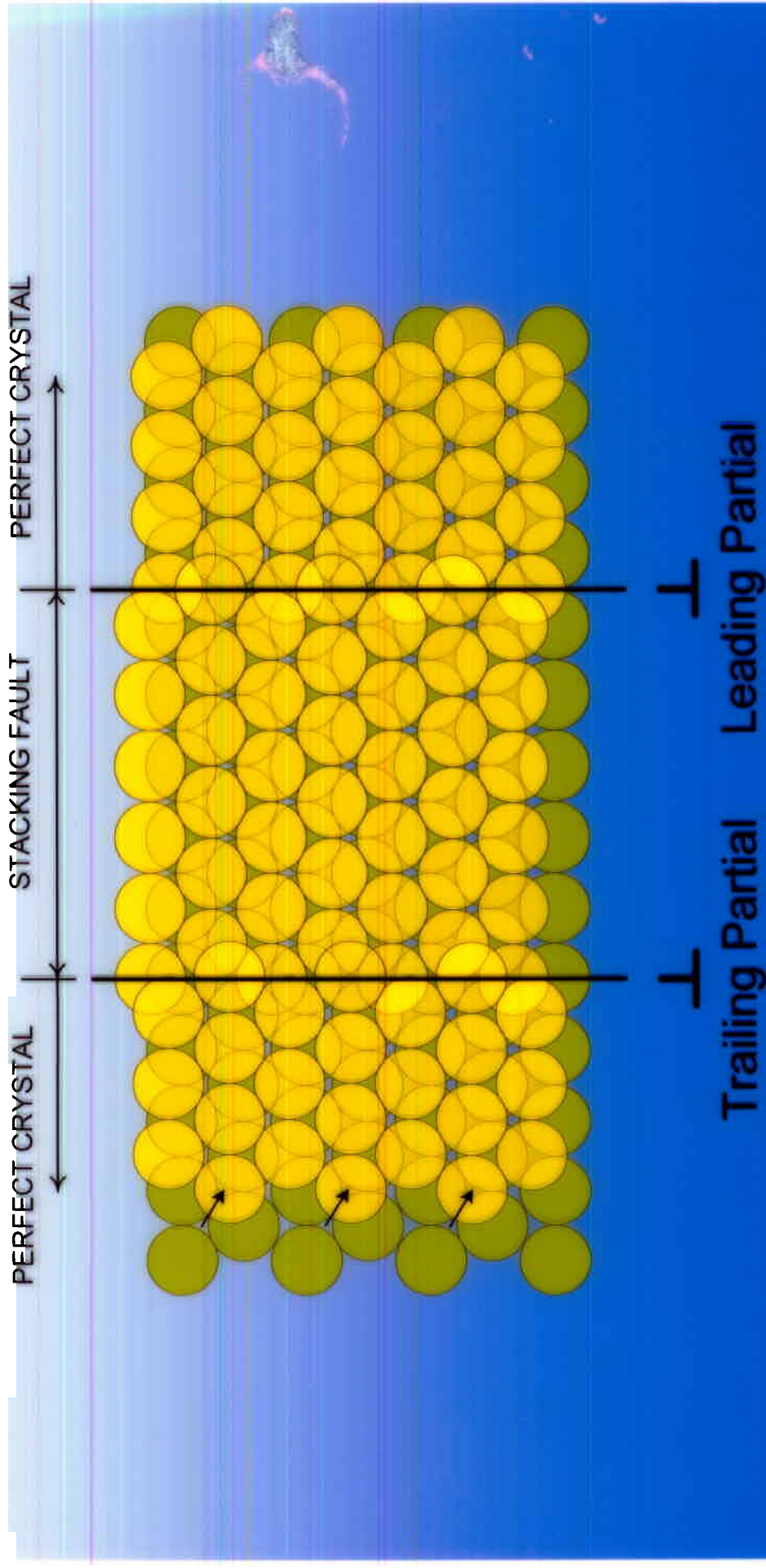


# Partial dislocations & stacking fault

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# Formation of an Extended Dislocation



# Two important themes

## 1. Geometry

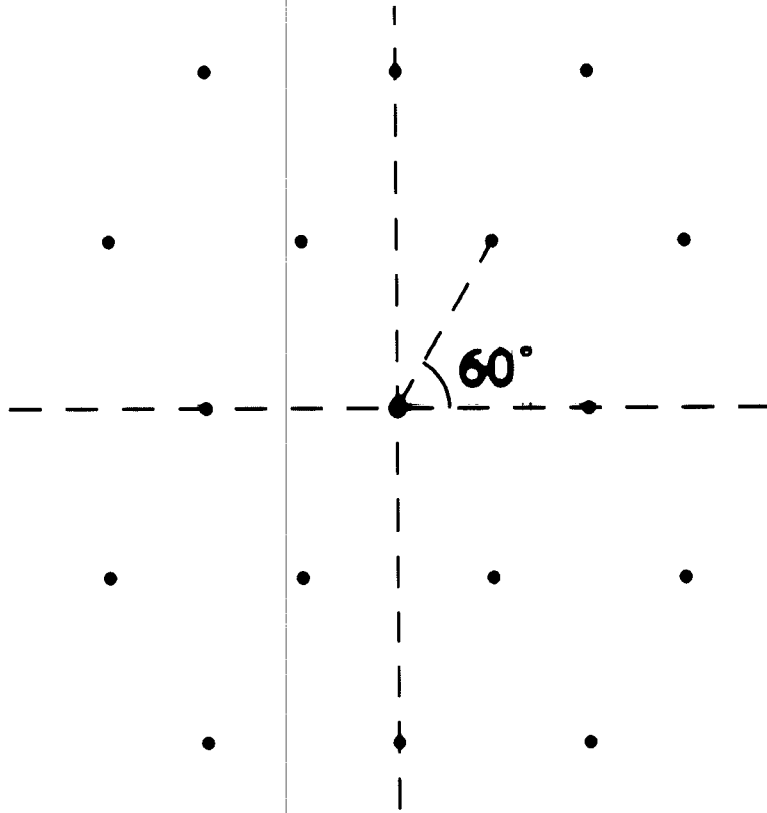
- **Crystal lattice:** Cubic, BCC, FCC, HCP, Diamond, ...
- **Lattice defects:** stacking fault, dislocations

## 2. Physics

- (a) **Why selecting the particular lattice?** (Quantum packing problem)
- (b) **Origin of cohesion: Why do atoms stick together?**
  - i. **Molecular** (van der Waals force)
  - ii. **Covalently-bonded** (e.g. Si)
  - iii. **Metallic**
  - iv. **Ionic** (e.g. NaCl)
  - v. **Hydrogen-bonded**

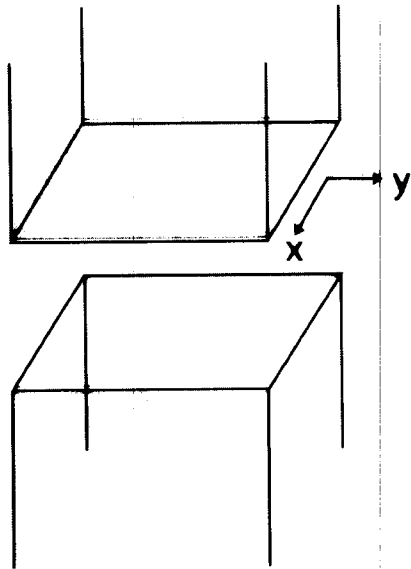






**Lattice L**

## Gamma surface energy



$\Gamma(x, y)$  = energy per unit cell at the surface when the two half crystals are displaced uniformly by  $(x, y)$

Example: FCC lattice in (111) direction

Illustration: displacing two triangular lattices

$$\Gamma(x, y) = \sum_{i \in L_1, j \in L_2} V_0(r'_i - r_j), \quad r'_i = r_i + (x, y)$$

Observation:  $\Gamma$  has the symmetry of the lattice

$$G = \text{point group of triangular lattice} \\ = \{I, R_{60}, R_x, R_y, \dots\}$$

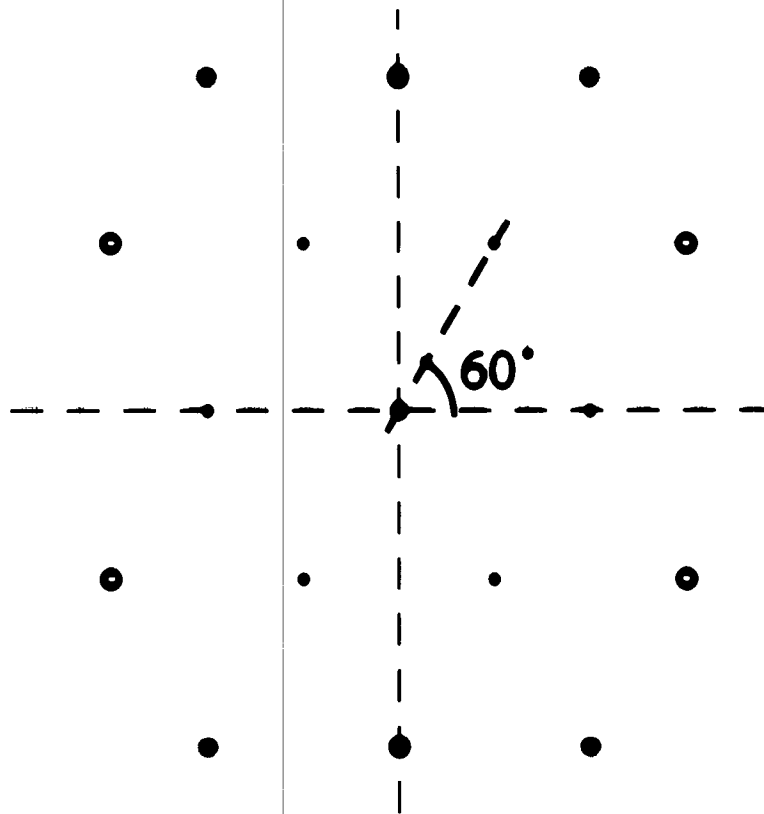
$L^*$  = dual lattice (Fourier space)  
= reciprocal lattice = triangular lattice

$$\Gamma(x, y) = \sum_{\text{orbits}} c_K O_K(x, y)$$

$\{O_K\}$  are invariant under  $G$

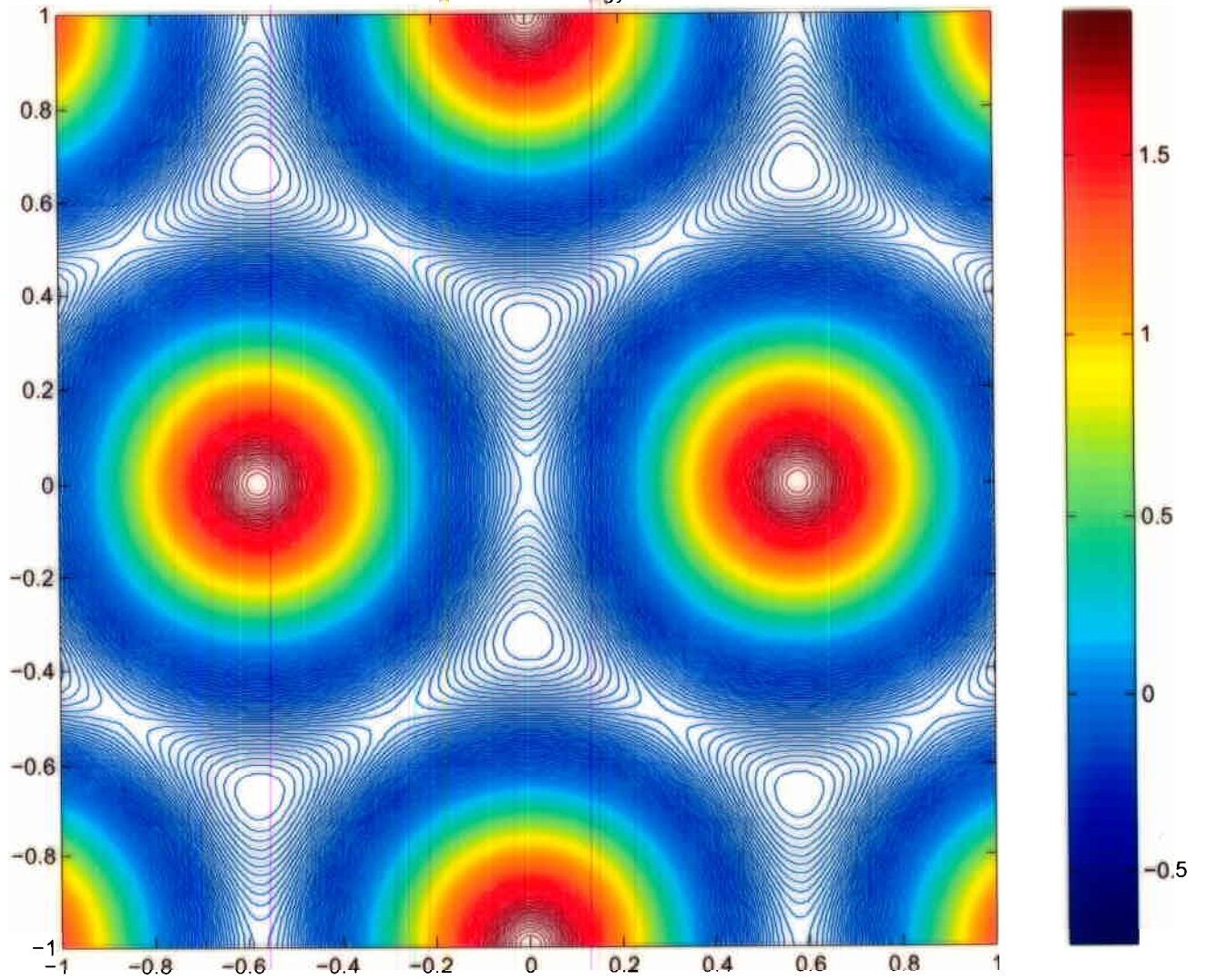
Leading order truncations give good approximation to  $\Gamma$ .

Conclusion: Geometry "determines"  $\Gamma$ .

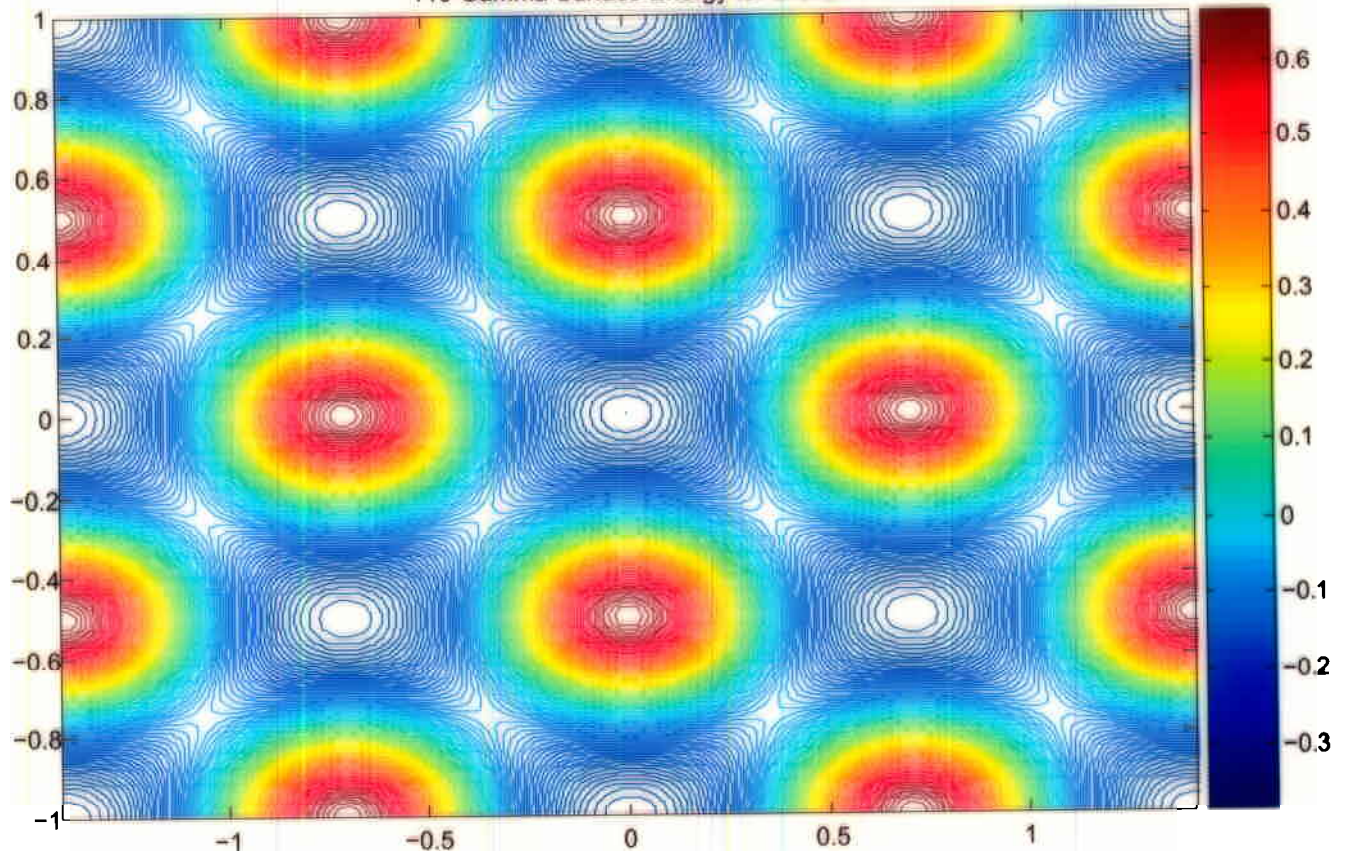


**Lattice L**

111 Gamma Surface Energy for F.C.C.



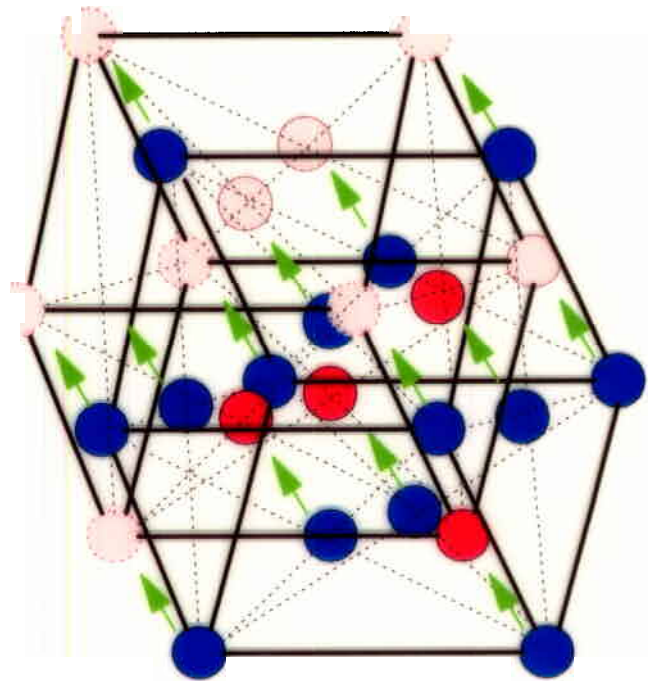
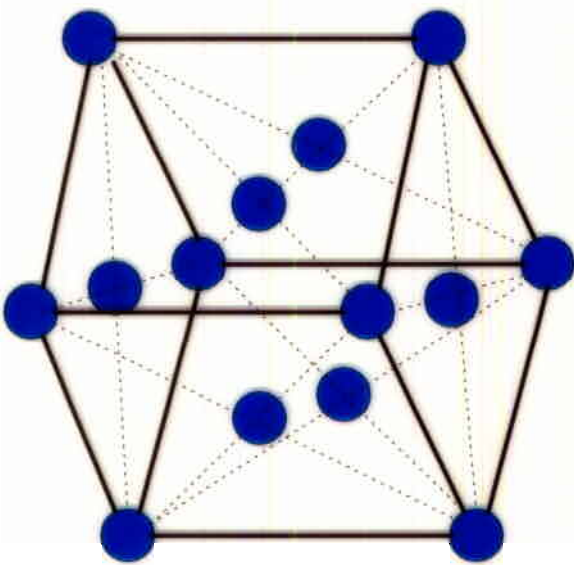
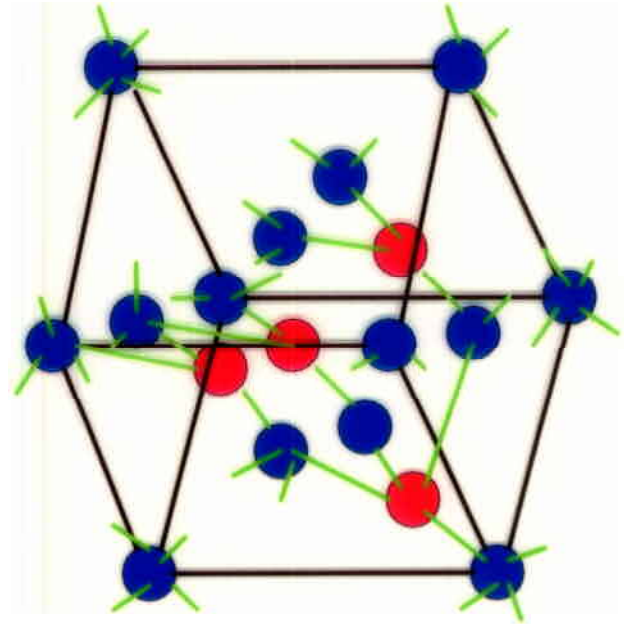
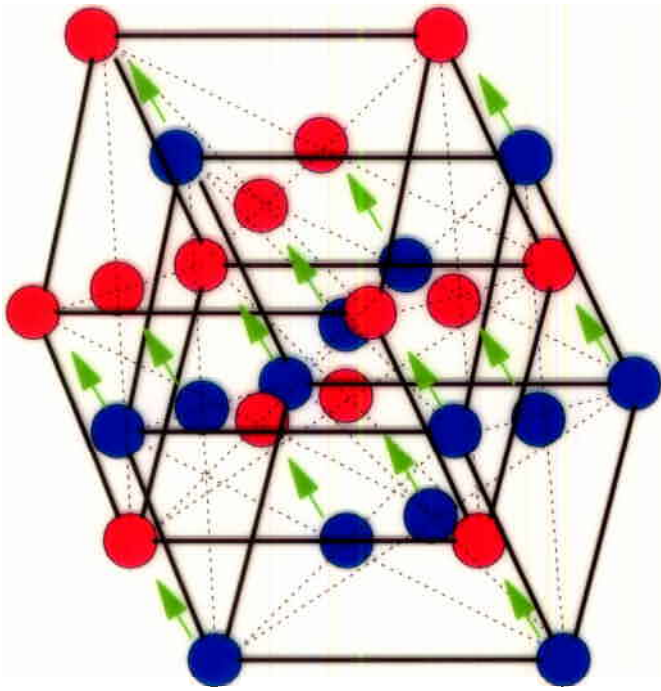
110 Gamma Surface Energy for B.C.C.



## **What questions do we need to address?**

- 1. Quantum packing problem:  
Why selecting the specific crystal structure?**
- 2. Stability of crystals (nonlinear elastic regime)**
- 3. Instability of crystals, defect formation**
- 4. Defect structure and dynamics (isolated)**
- 5. Defect-defect interactions, plasticity**







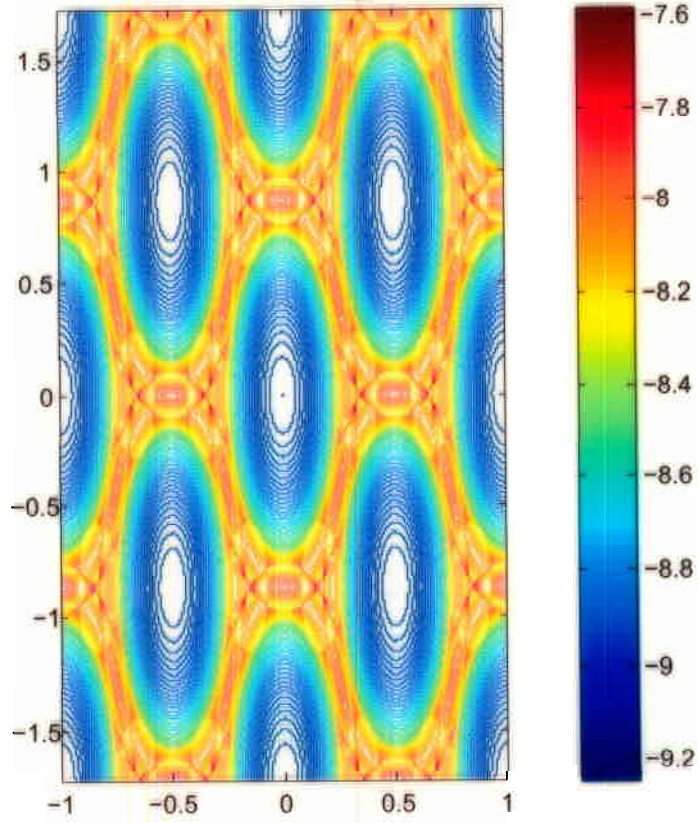
## **Diamond lattice**

$\Gamma(x, y)$  nearly singular

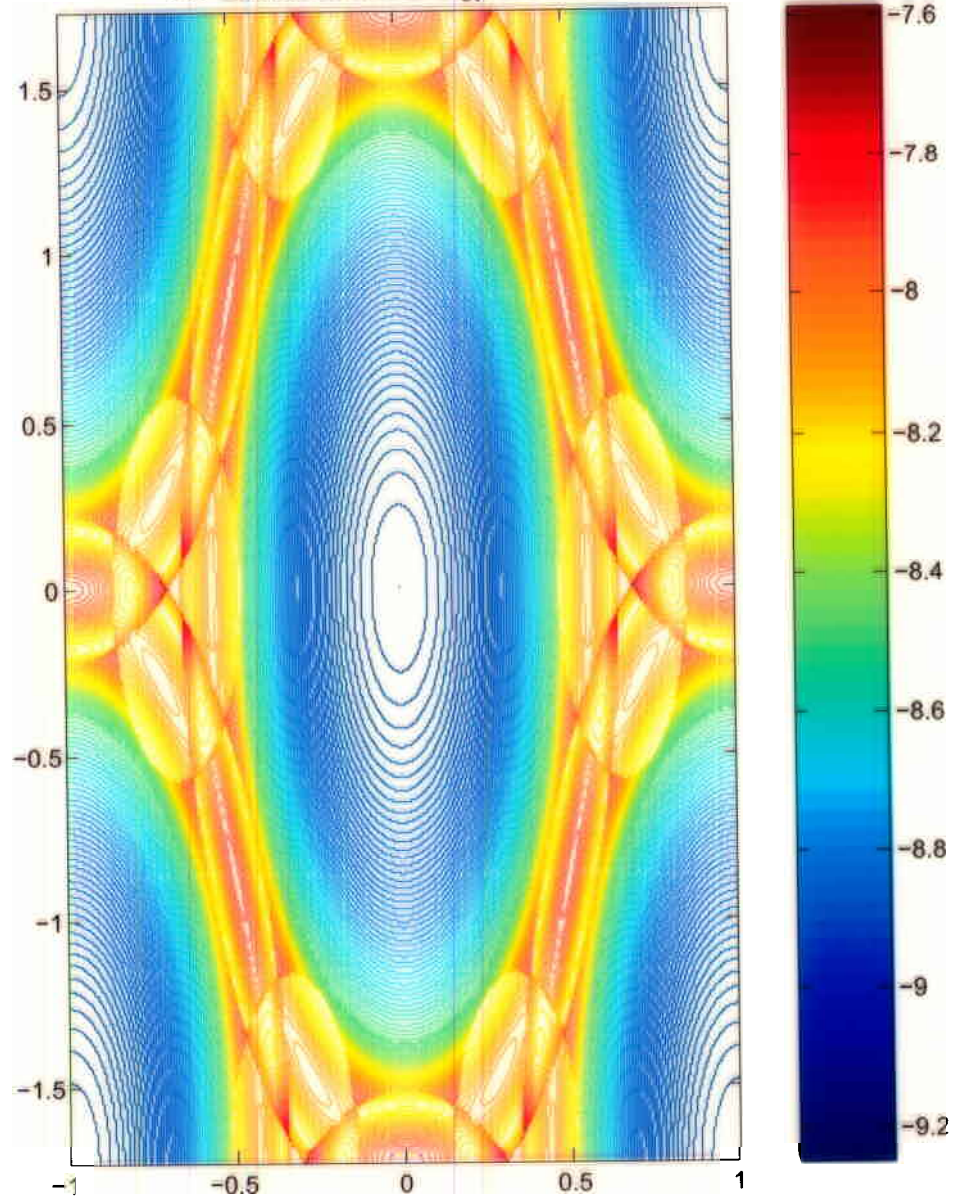
Geometry tells little.

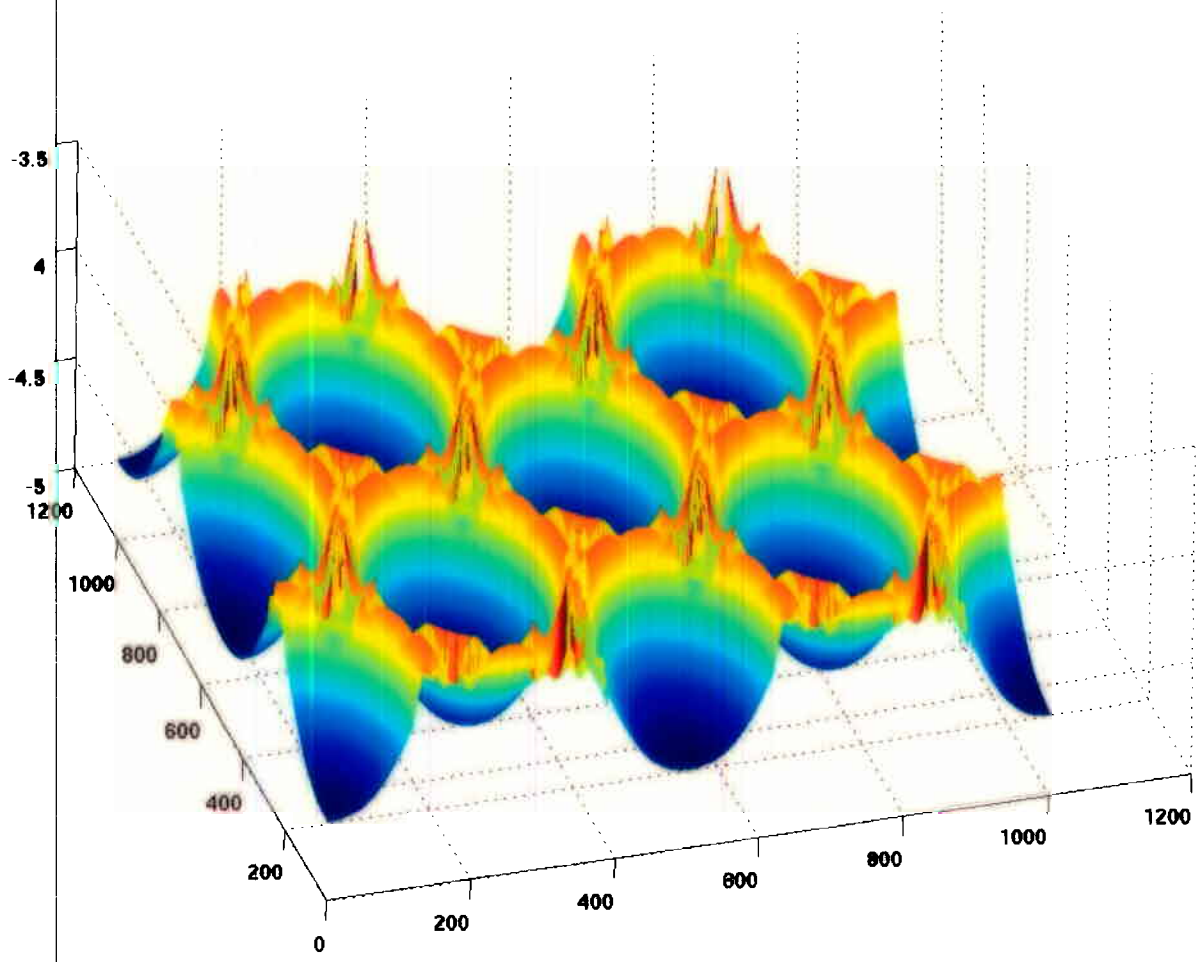
Physics (details of the atomic interaction) is important!

111 Gamma Surface Energy of Diamond



111 Gamma Surface Energy for Diamond

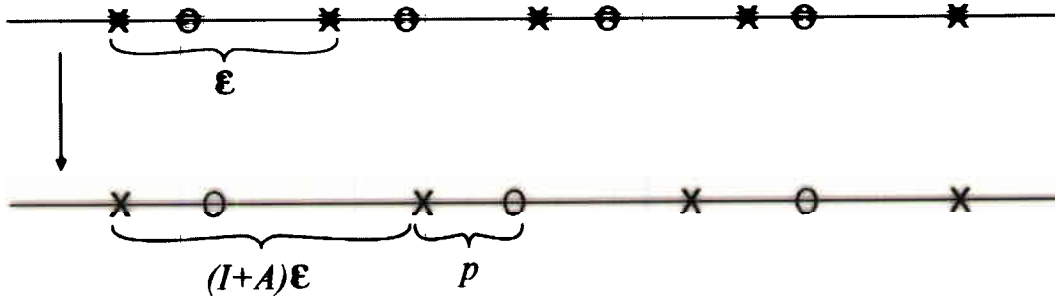




# Nonlinear elastic regime

(E and Ming)

## Cauchy-Born rule



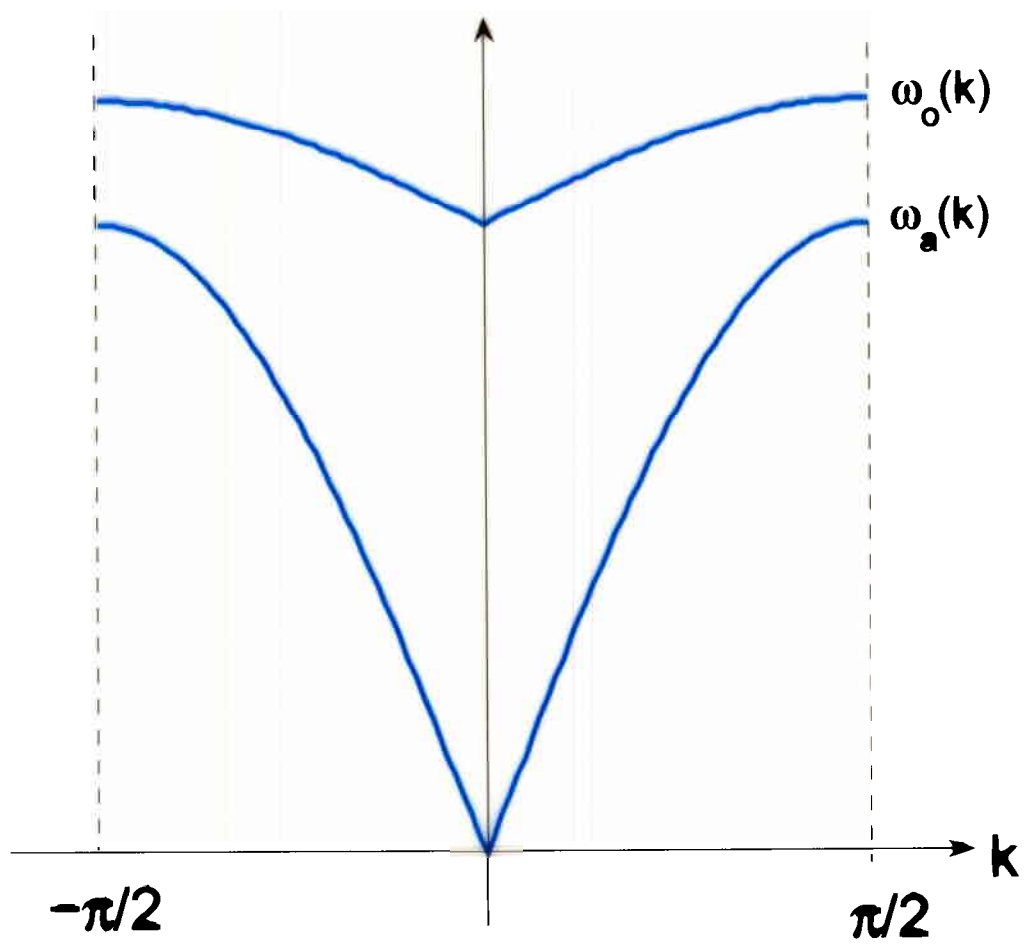
$W(A, p)$  = energy of unit cell in the deformed, (by  $A$ ), and displaced (by  $p$ ) configuration.

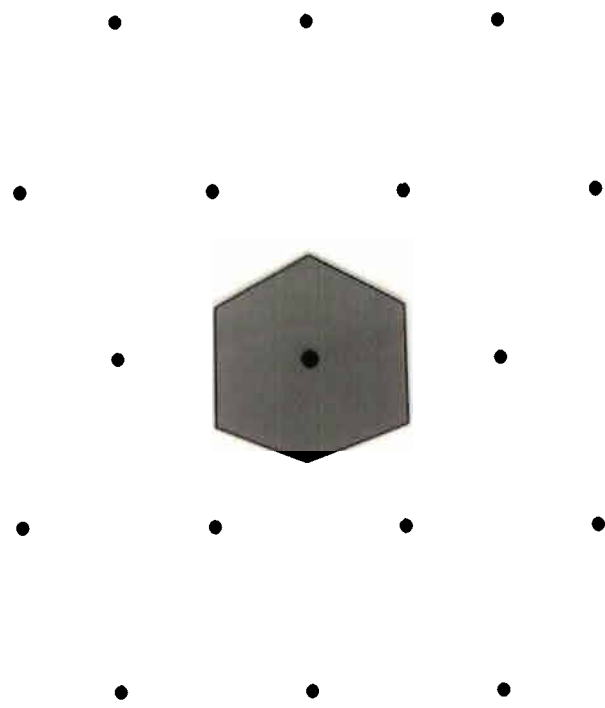
$$W_{CB}(A) = \min_p W(A, p)$$

$$I(v) = \int_{\Omega} \{W_{CB}(\nabla v) - f(x)v(x)\} dx \quad (*)$$

$$I_h(\{v_j\}) = V(y_1, \dots, y_N) - \sum f(x_j)v_j \quad (**)$$

$$y_j = x_j + v_j$$





**Assumption A:**

$$\begin{pmatrix} D_{AA}W(A, p), & D_{Ap}W(A, p) \\ D_{Ap}W(A, p), & D_{pp}W(A, p) \end{pmatrix} > 0 \quad \begin{array}{l} A = 0, \\ p = p_0 \end{array}$$

**Assumption B: "phonons"**

$$\omega_a(k) \geq \Lambda_1 |k|$$

$$\omega_0(k) \geq \frac{\Lambda_2}{\varepsilon}$$

$k \in$  1st Brillouin zone

**Theorem:** ( $p \geq d =$  dimension)  $\exists K, R$  such that if  $\|f\|_{L^p} \leq K$ , then  $\exists$  unique solution of (\*),  $v = u_{CB}$ , such that

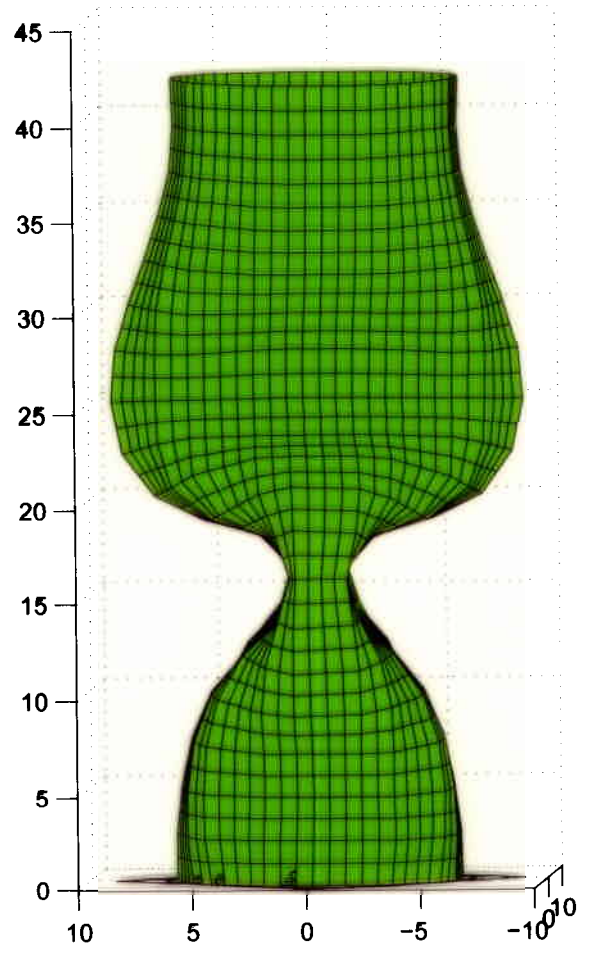
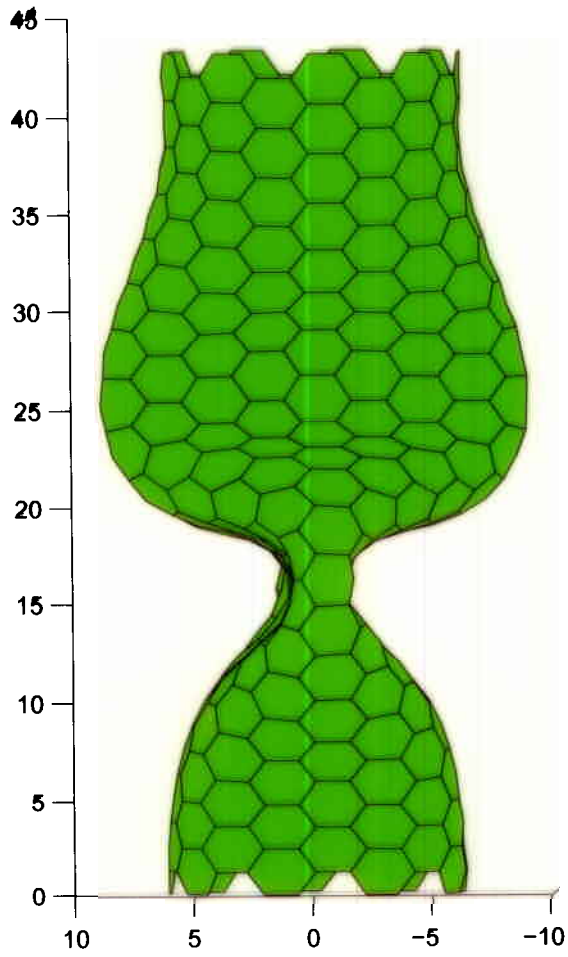
$$\|u_{CB}\|_{W^{2,p}} \leq R$$

**Theorem:**  $\exists$  solution  $y$  of (\*\*), such that

$$\|y - y_{CB}\|_1 \leq C\varepsilon$$

$y_{CB}(x) = x + u_{CB}(x)$ ,  $\|\cdot\|_1 =$  discrete  $H^1$  norm





## Remarks

1. **Result is sharp**

**Classification of instability, if  $\|f\|_{L^p} > K$   
Defect formation.**

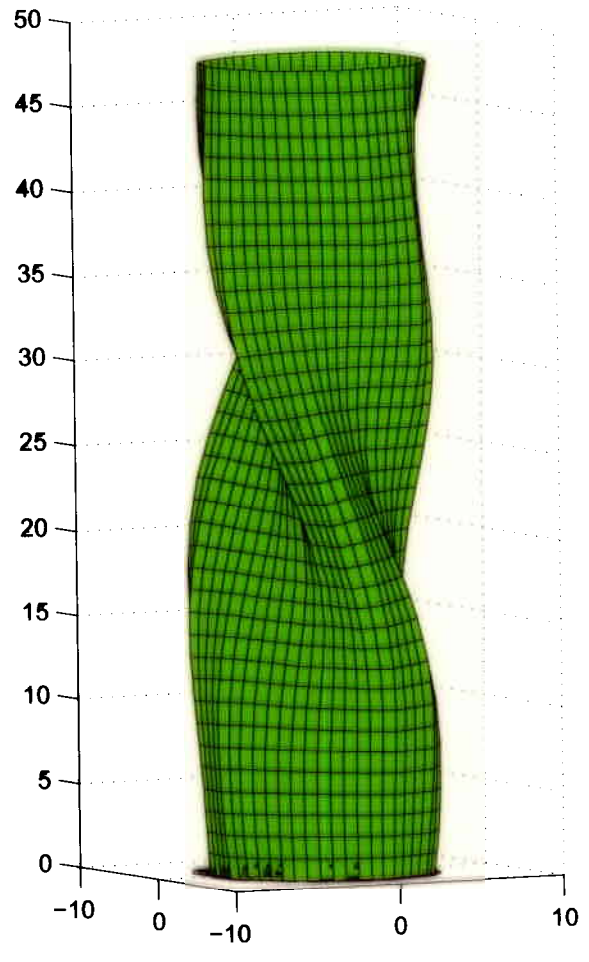
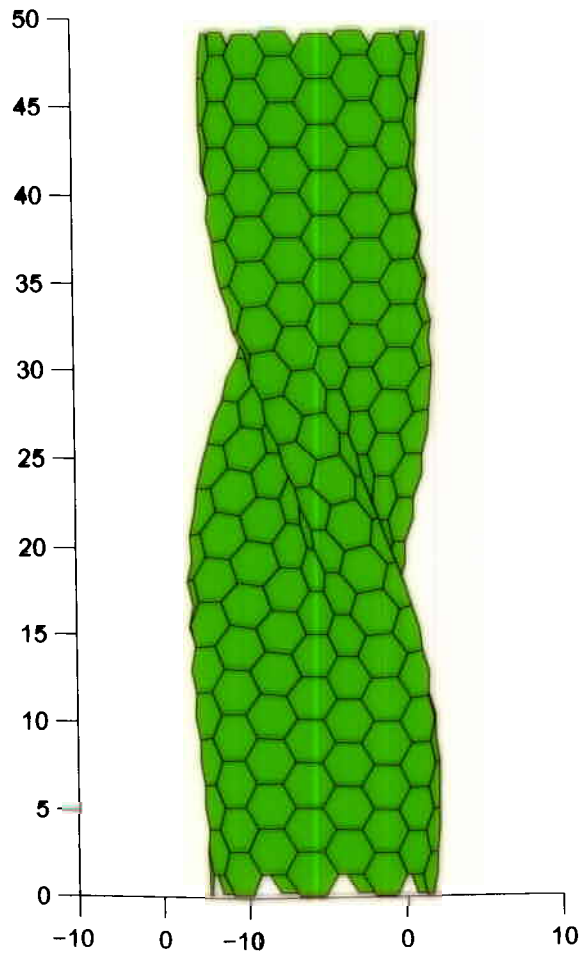
2. **Result can be extended to dynamic problems.**

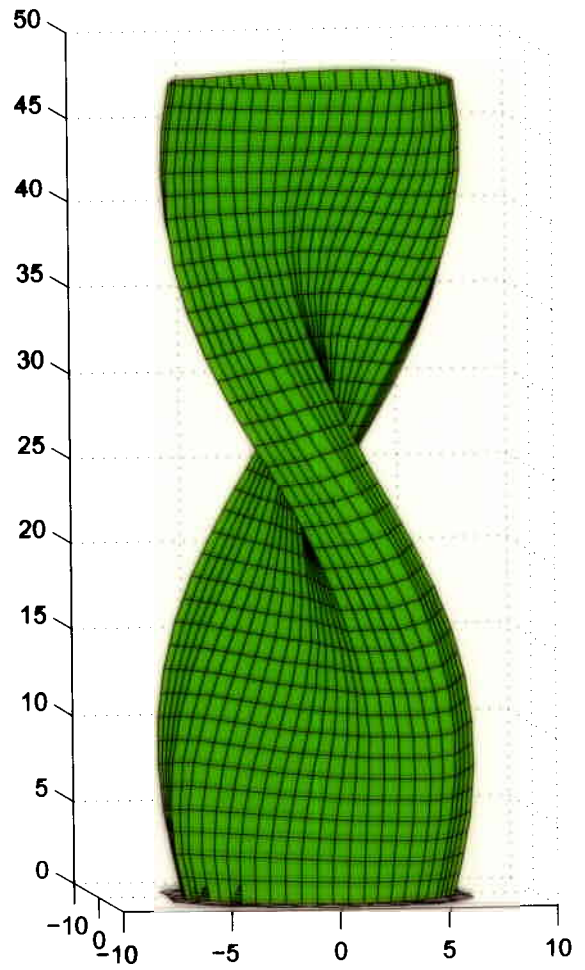
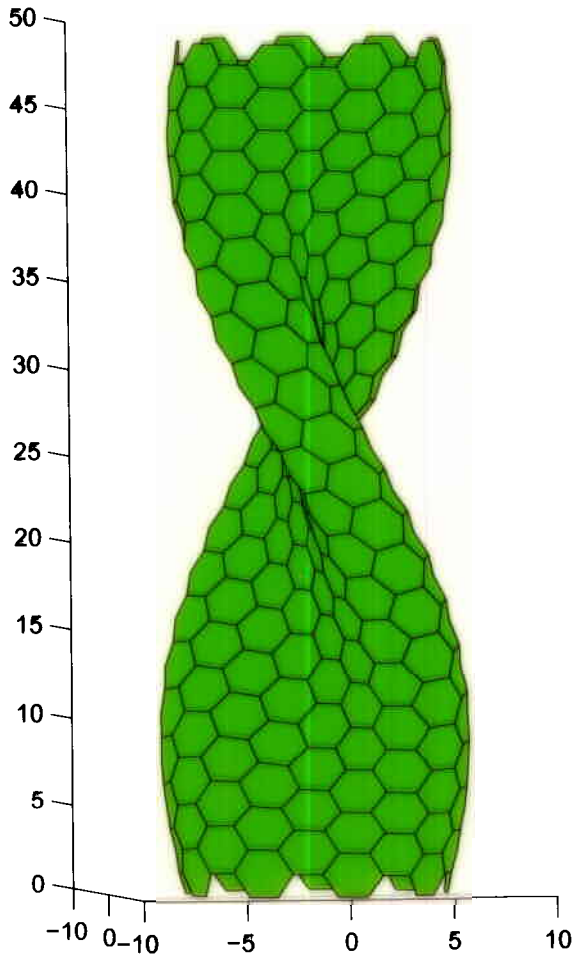
3. **Previous work of Blanc, LeBris and P.L. Lions. Assume  $y_j = x_j + u(x_j)$ ,  $u$  is smooth, and study limit of  $V$ .**

**Example: Lennard-Jones potential, next nearest neighbor interaction**

**Case 1. Triangular lattice**

**Case 2. Square lattice**





## **Summary**

- 1. Atomistically-based mathematical theory of solids is both important and possible.**
- 2. Geometry and physics themes helpful.**

**Reason:** Solids are much more *multi-scaled*  
(and heterogeneous)

Atomic scale: Atoms, electrons  
    → crystal lattice  
    → lattice defects

Macroscopic: randomly packed grains

**Hierarchy of structures**

